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Comparing spherical and irregular powders in the selective laser melting of Ti-53%Nb alloy

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29 slides.*

A série “Comunicação Técnica” compreende trabalhos elaborados por técnicos do IPT, apresentados em eventos, publicados em revistas especializadas ou quando seu conteúdo apresentar relevância pública.

COMPARING SPHERICAL AND IRREGULAR POWDERS IN THE SELECTIVE LASER MELTING OF Ti-53%Nb ALLOY

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Objective of the project

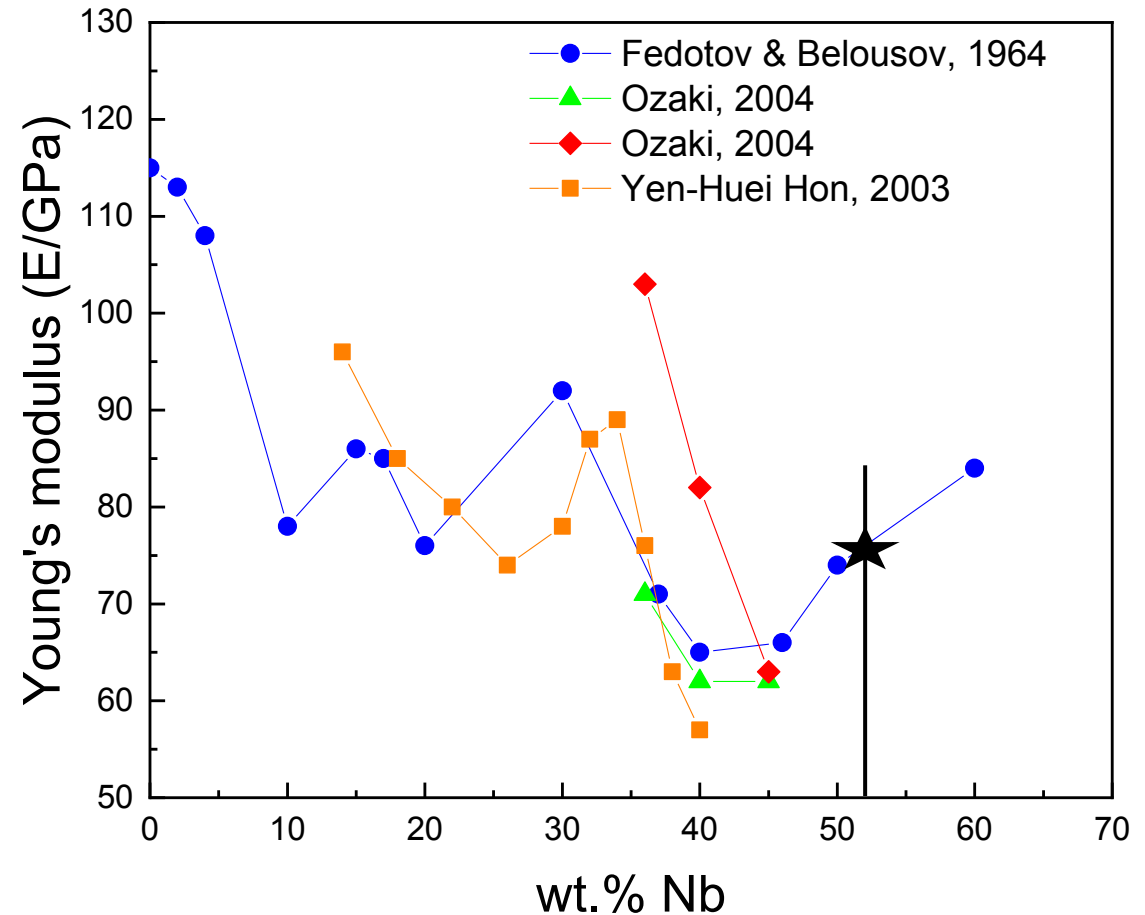
- To evaluate the possibility of producing orthopaedic Implants of Ti-53%Nb alloy by Selective Laser Melting
 - The Niobium Company (CBMM) wants to promote the use of Niobium in new applications
 - Niobium, like titanium, promotes osteointegration
 - Orthopedic implants work better when their elastic modulus is similar to that of the bones
 - Niobium addition to titanium lowers the elastic modulus

Young's modulus Ti–Nb

Niobium decreases the elastic module of titanium

There is some scatter about the effect of niobium on the elastic constant

The Ti-53%NB was chosen because of alloy commercial availability: it is a superconductor composition.



(7) C. Schulze, M. Weinmann, C. Schweigel, O. Keßler, R. Bader, Mechanical Properties of a Newly Additive Manufactured Implant Material Based on Ti-42Nb, Materials (Basel). 11 (2018) 124. doi:10.3390/ma11010124.

Niobium – Titanium system

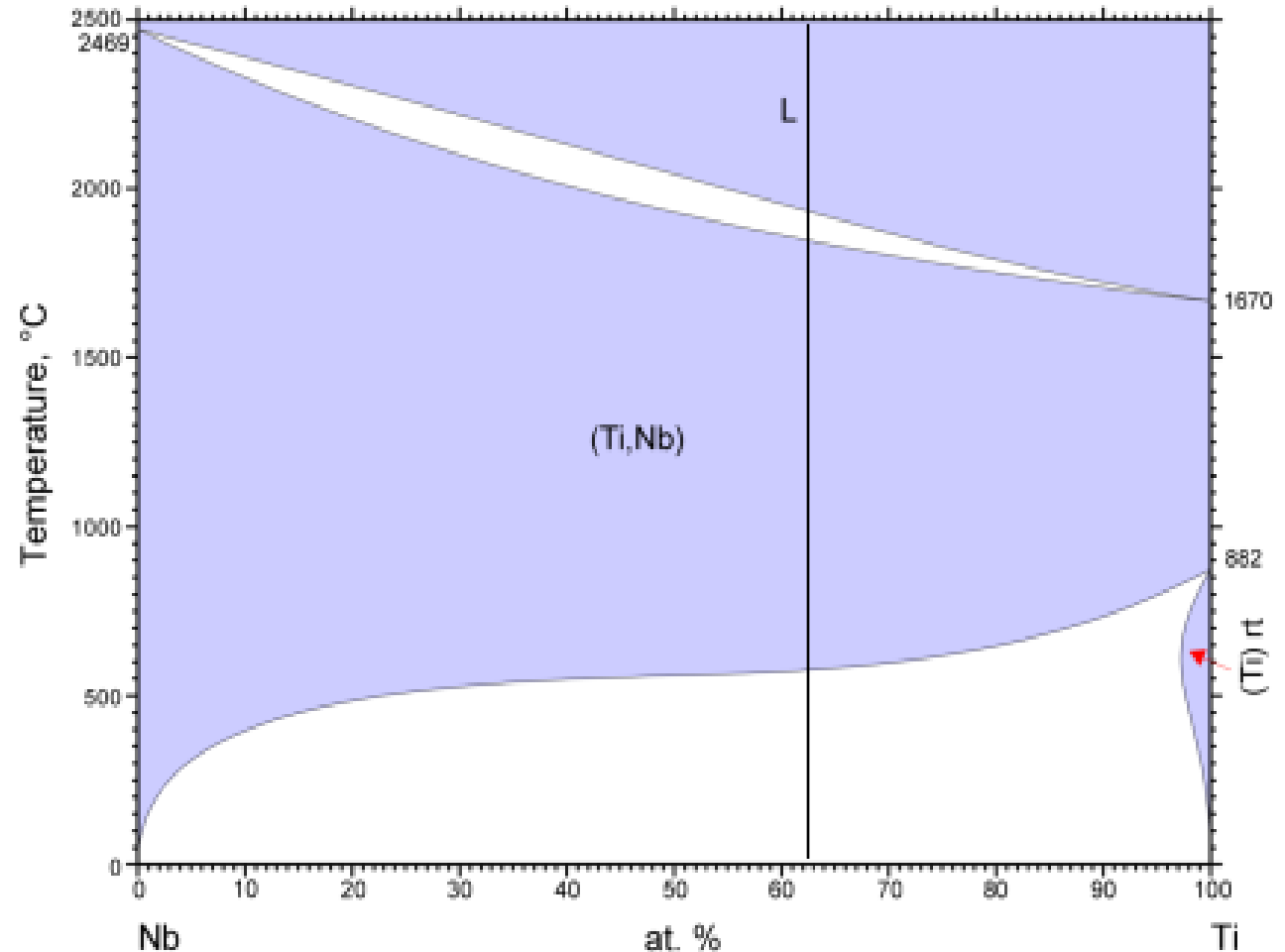
Phases of the system Ti – Nb		
Phase	Crystal structure	Composition at%. Nb
α	HCP	0-2.5
β	BCC	0-100

Although the phase diagram predicts a $\alpha + \beta$ structure, beta is strongly metastable.

(7) M. Hansen, E.L. Kamen, H.D. Kessler, D.J. McPherson, Systems Titanium-Molybdenum and Titanium-Columbium, JOM. 3 (1951) 881–888. doi:10.1007/BF03397396.

(8) J.L. Murray, The Nb–Ti (Niobium-Titanium) system, Bull. Alloy Phase Diagrams. 2 (1981) 55–61. doi:10.1007/BF02873704.

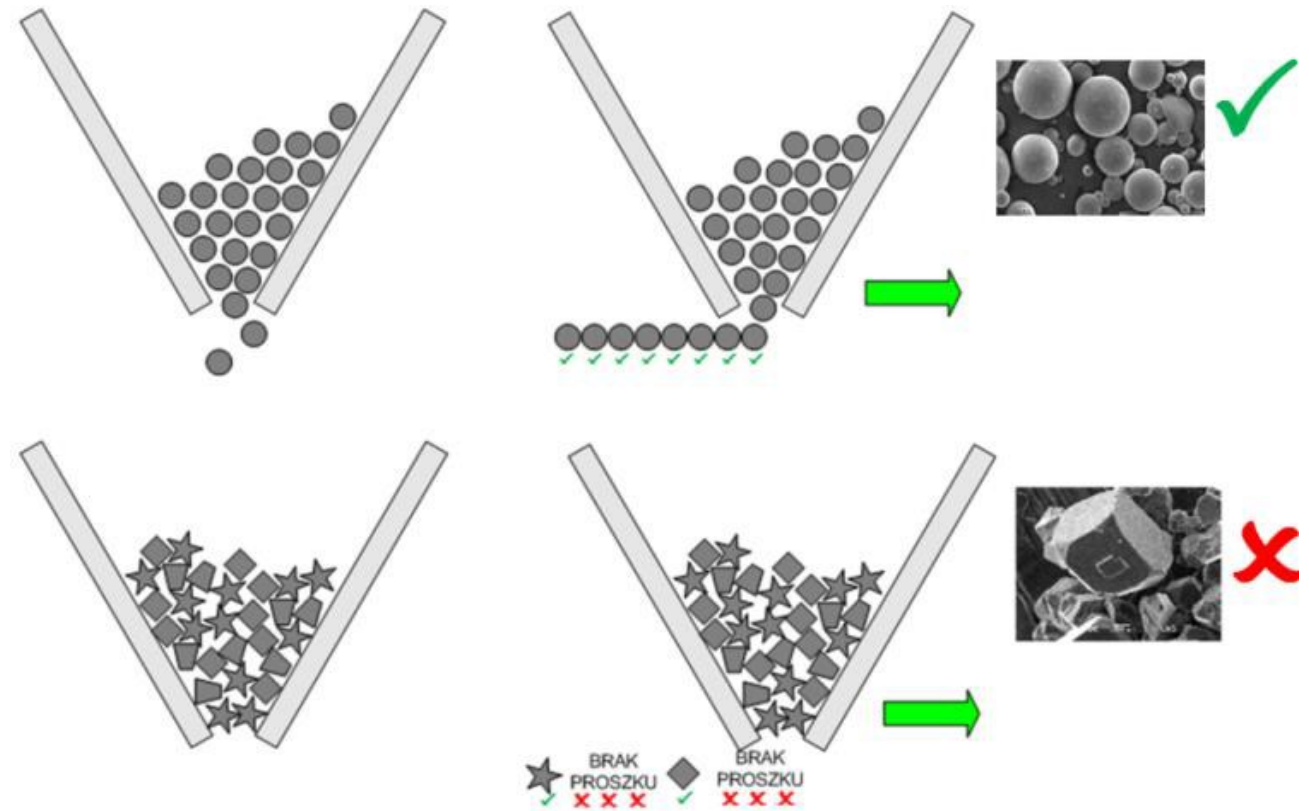
(9) C.M. Lee, C.P. Ju, J.H. Chern Lin, Structure-property relationship of cast Ti-Nb alloys., J. Oral Rehabil. 29 (2002) 314–22. <http://www.ncbi.nlm.nih.gov/pubmed/11966963>.



Powders

Cost and flowability

- Spherical shaped powder is the common solution for Ti (U\$ 333/kg)
- Ti irregular shaped powder may be a better solution (U\$ 50/kg ?)
- Apparent density and flowability may be impaired.



Layer deposition: a) spherical powder, b) irregular powder (6).

(5) W. Xu, S. Xiao, X. Lu, G. Chen, C. Liu, X. Qu, Fabrication of commercial pure Ti by selective laser melting using hydride-dehydride titanium powders treated by ball milling, *J. Mater. Sci. Technol.* 35 (2019) 322–327. doi:10.1016/j.jmst.2018.09.058.

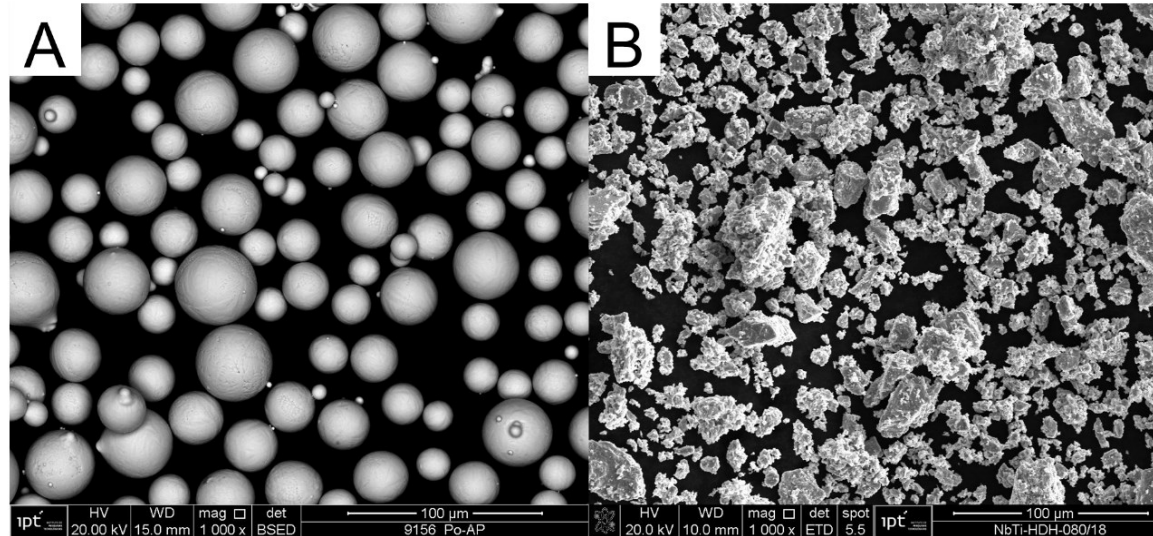
(6) T. Kurzynowski, E. Chlebus, B. Kuźnicka, J. Reiner, Parameters in selective laser melting for processing metallic powders, in: E. Beyer, T. Morris (Eds.), 2012: p. 823914. doi:10.1117/12.907292.

Objective of this paper

- To evaluate the viability of obtaining high density parts using HDH powder

Materials

AP spherical shaped powder



HDH irregular shaped powder

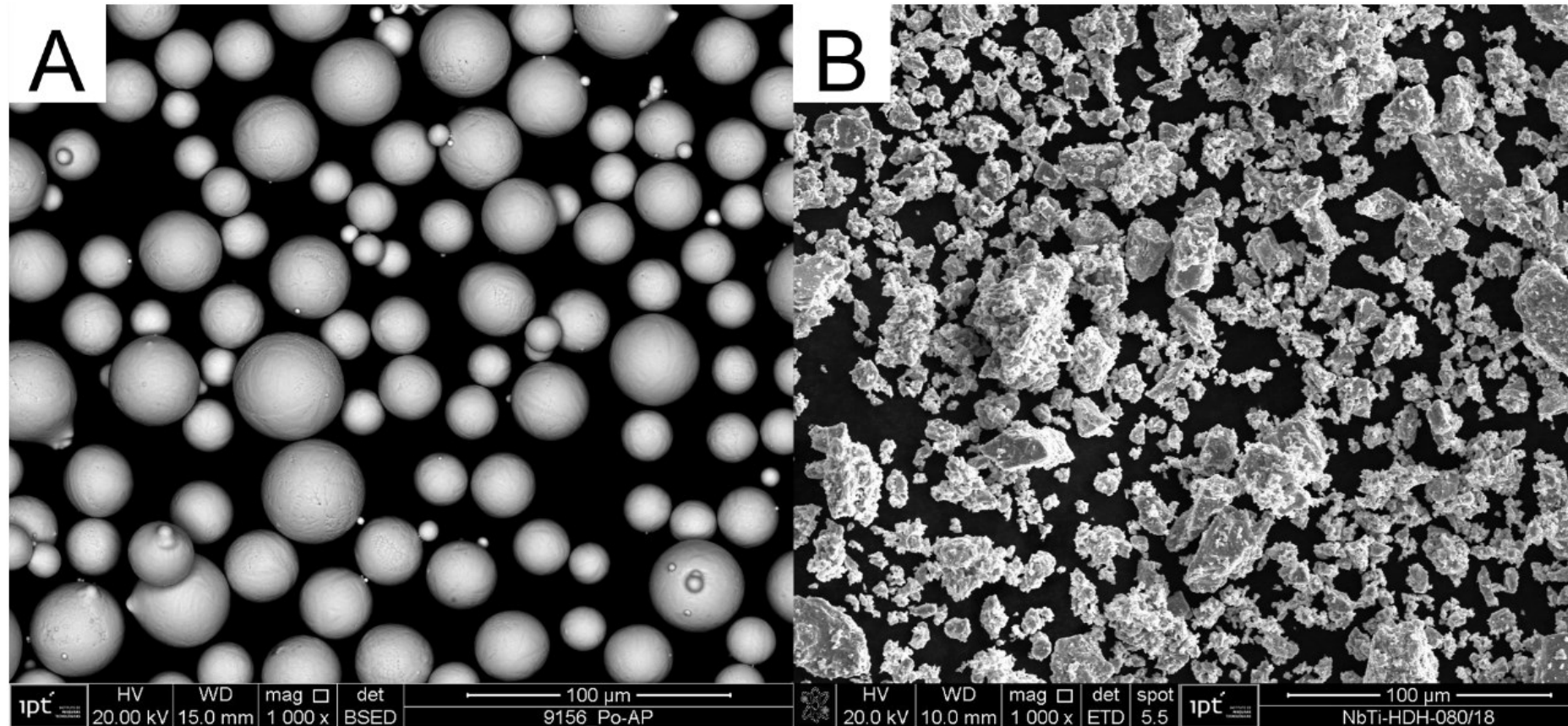
Powder characteristics.

Powder process	Plasma Atomization	Hydrogenation-Dehydrogenation
Morphology	Spherical	Irregular
ρ_{DRX} (g/cm ³)	6.059	6.046
ρ_a (mg/mm ³)	3.49	1.85
O initial (ppm)	1200	3400
N initial (ppm)	209	430
D10, D50, D90	19 μ m, 31 μ m, 51 μ m	12 μ m, 31 μ m, 67 μ m
Hall flow (s/50g)	20	No flowability

Materials

AP spherical shaped powder

HDH irregular shaped powder



SLM parameters

- As the apparent density of the HDH powder was 50% of the atomized powder, a double layer thickness was chosen, so that the mass energy density was similar.

	Atomized powder	HDH powder
Average particle size (μm)	31	31
Apparent density ρ_a (g/cm^3)	3.49	1.85
Layer thickness D_s (μm)	30	60

$$E_v = \frac{P_L}{V_S * h_S * D_S}$$

E_v is Volumetric energy density

$$E_m = \frac{E_v}{\rho_a}$$

E_m is mass energy density

SLM parameters

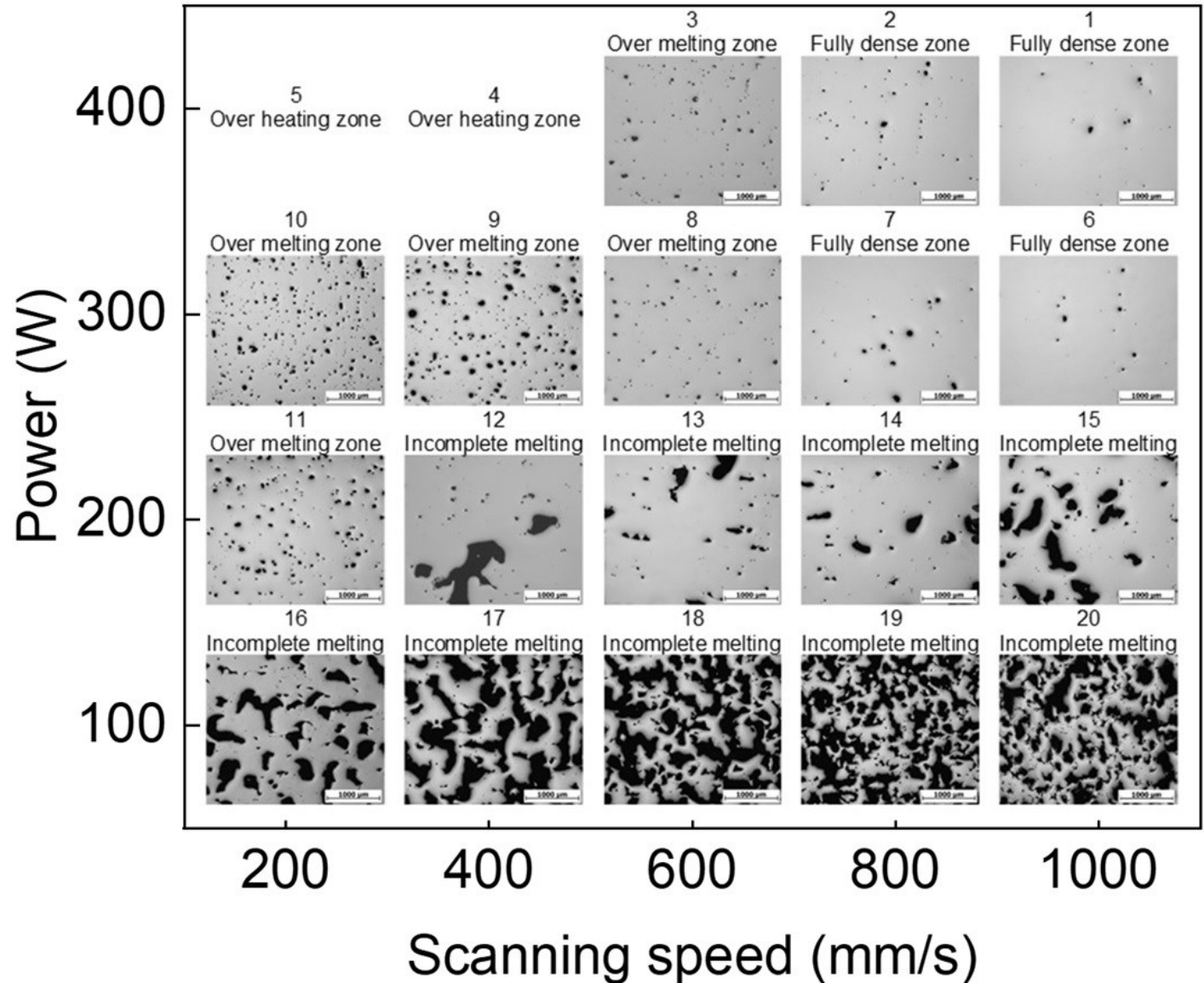
- Samples were made in a Concept Laser M2 machine, at ISI Laser Labs, Joinville, Brazil.
- For the HDH series, 20 conditions were investigated
 - Laser Power from 100 to 400W
 - Scanning speed from 200 to 1000 mm/s
 - Results suggested a better screen for the second trial
- For the atomized powder, 24 conditions were investigated
 - Laser power from 150 to 300W, from 900 to 1400mm/s

Results

Methods

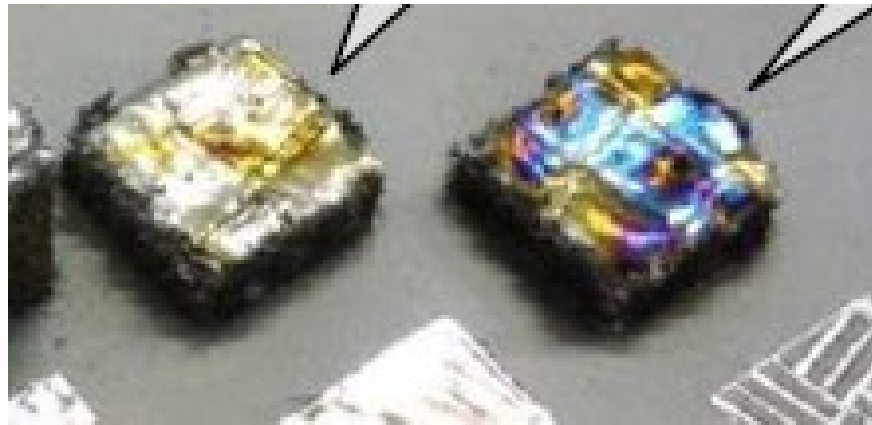
- Relative density was obtained by the Archimedes' principle (hydrostatic density) and pore area fraction with the Fiji Software.
- Microstructural analyses by optical microscopy and scanning electron microscopy.
- Microhardness testing using Vickers indentation with a load of 200 g.
- Measurements of oxygen and nitrogen were performed by the inert gas fusion-infrared absorption.

Influence of power and scanning speed

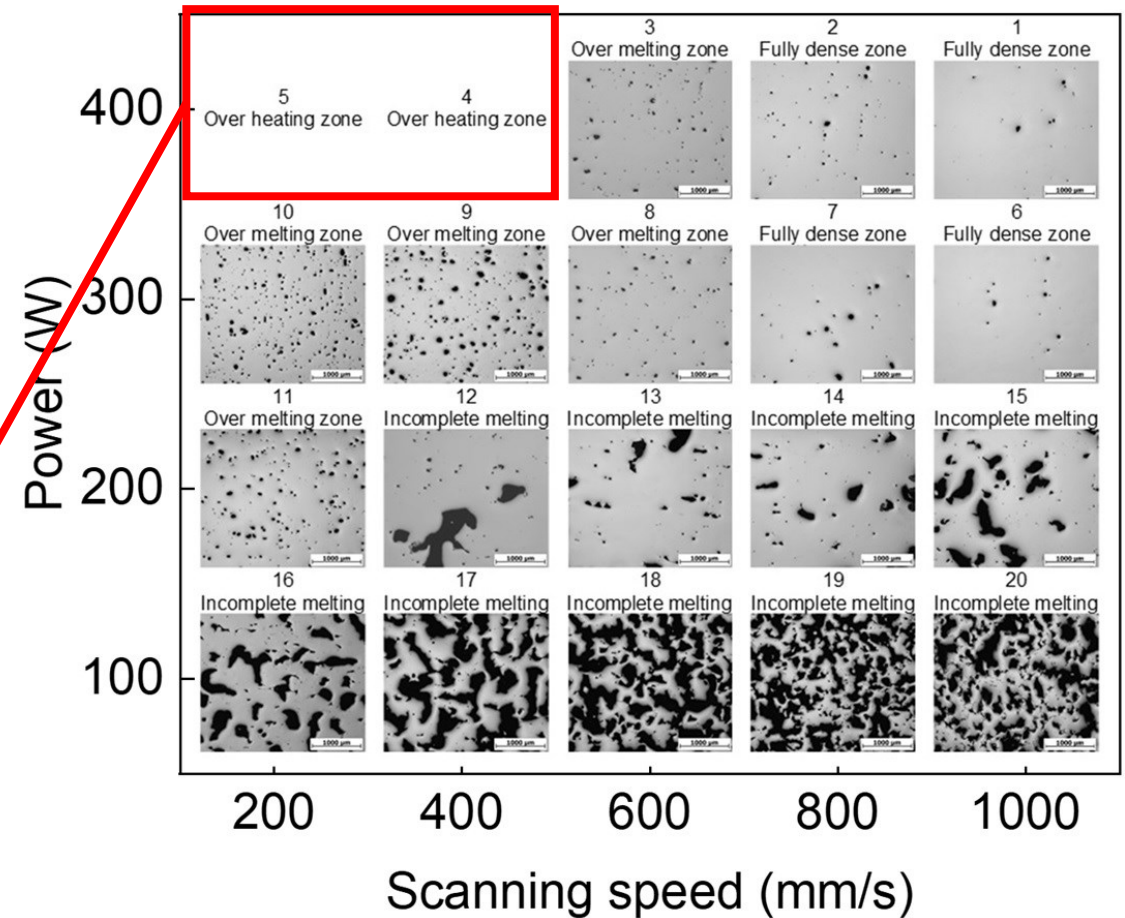


Over heating zone

High power and low scanning speed. Pieces are deformed due to high deposited energy, so it is impossible to manufacture final parts.



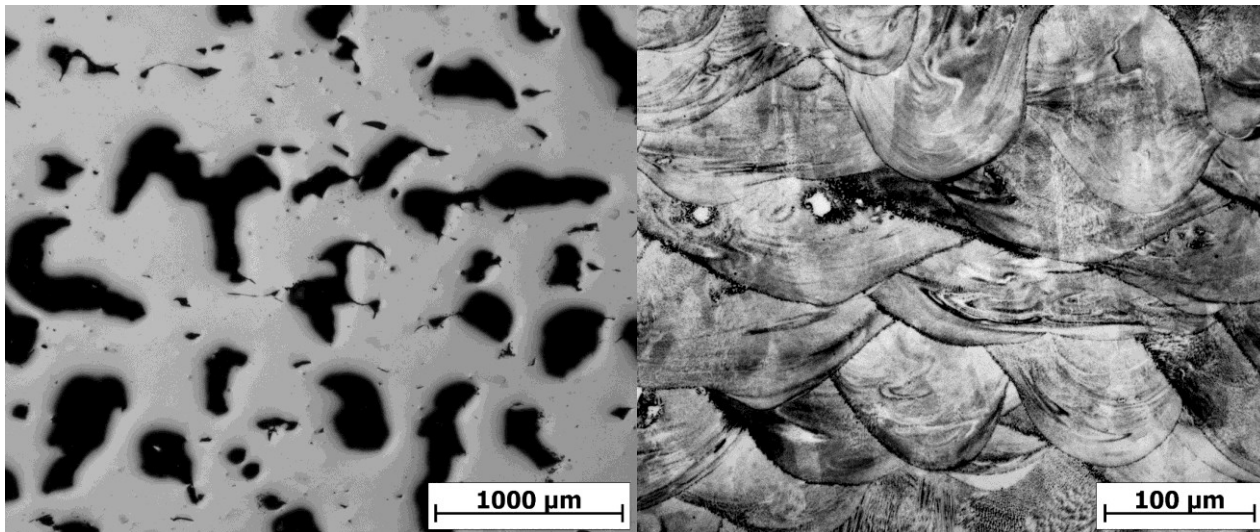
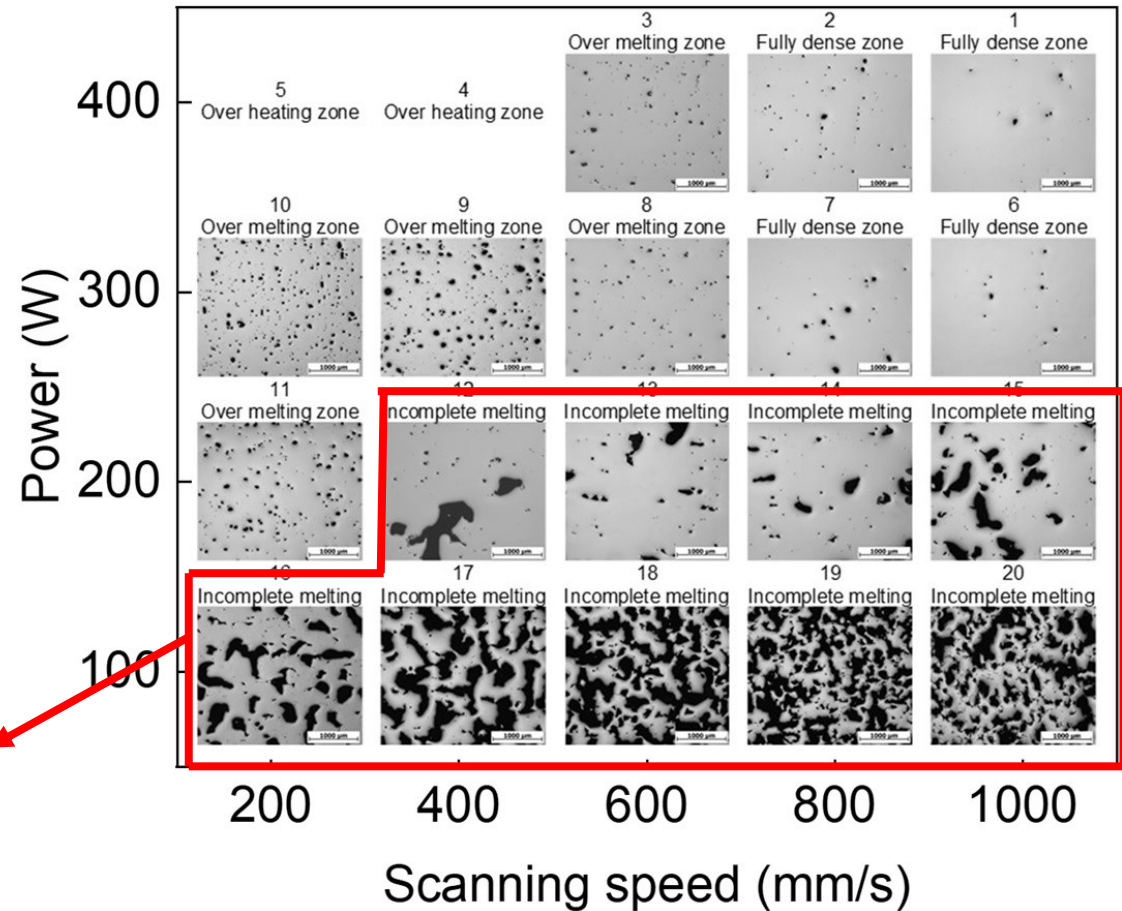
Samples from HDH powder



Incomplete melting zone

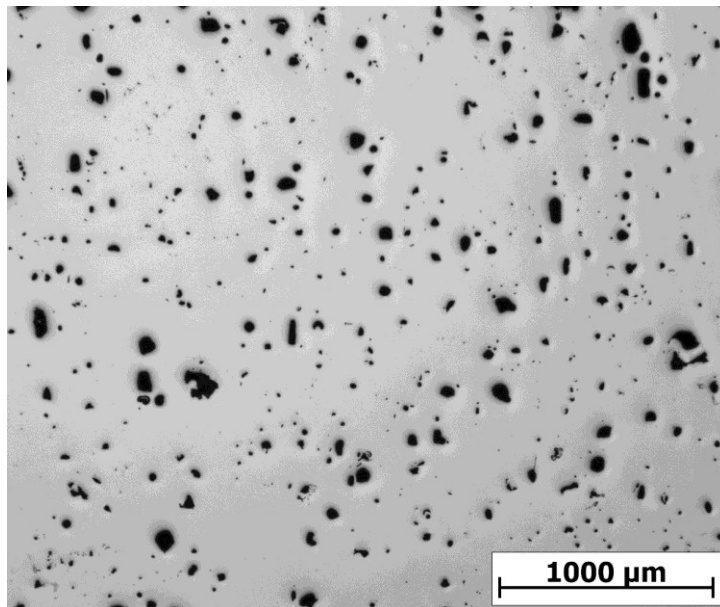
Samples from HDH powder

Low power and high scanning speed. It is possible to manufacture pieces with present of irregular porosity and high content of segregation. It is also known as lack of fusion.

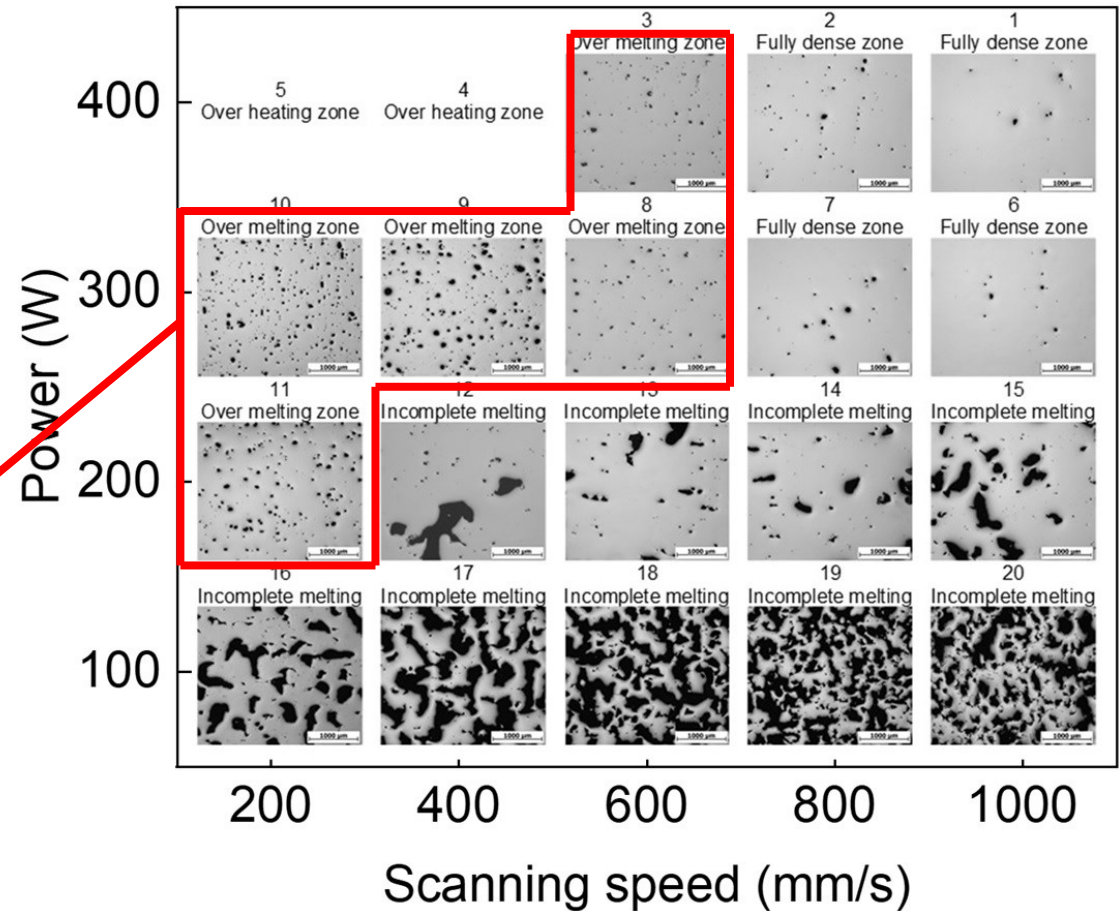


Over melting zone

High power (slightly less than in the first zone) and low scanning speed. It is possible to manufacture pieces with spherical porosity due to trapped gas.



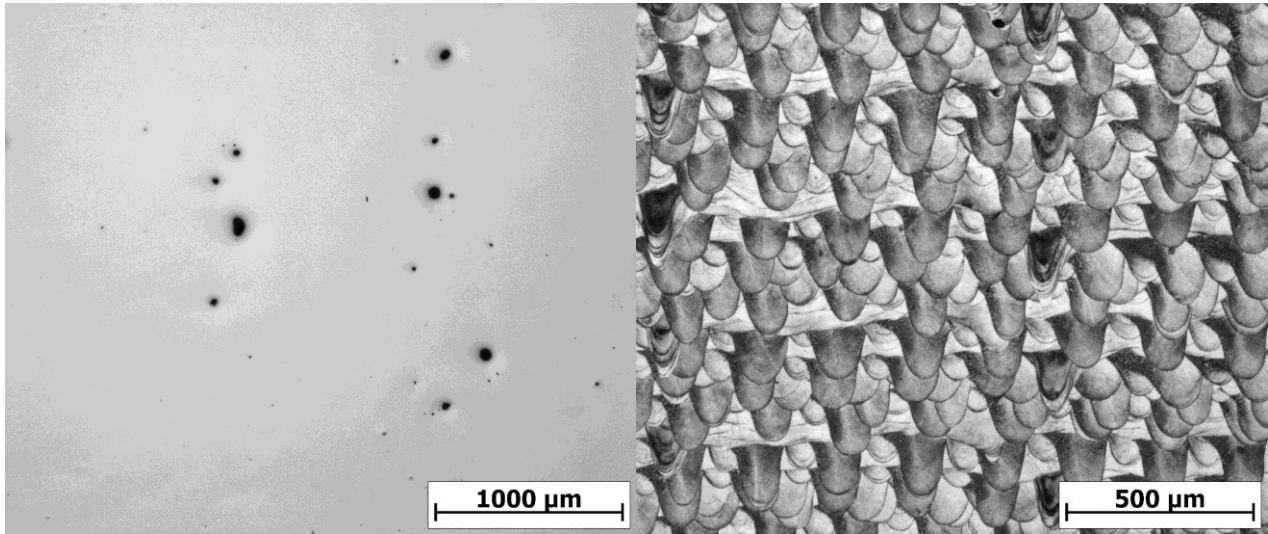
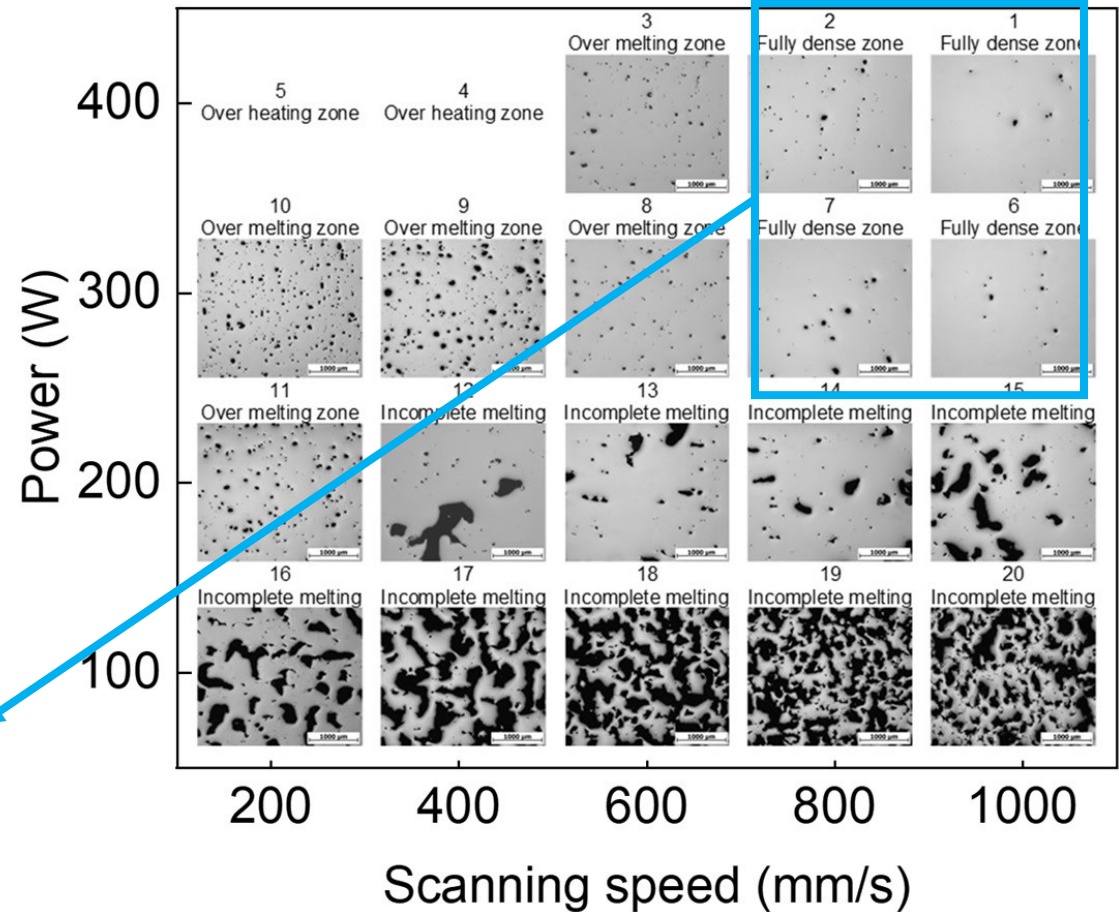
Samples from HDH powder



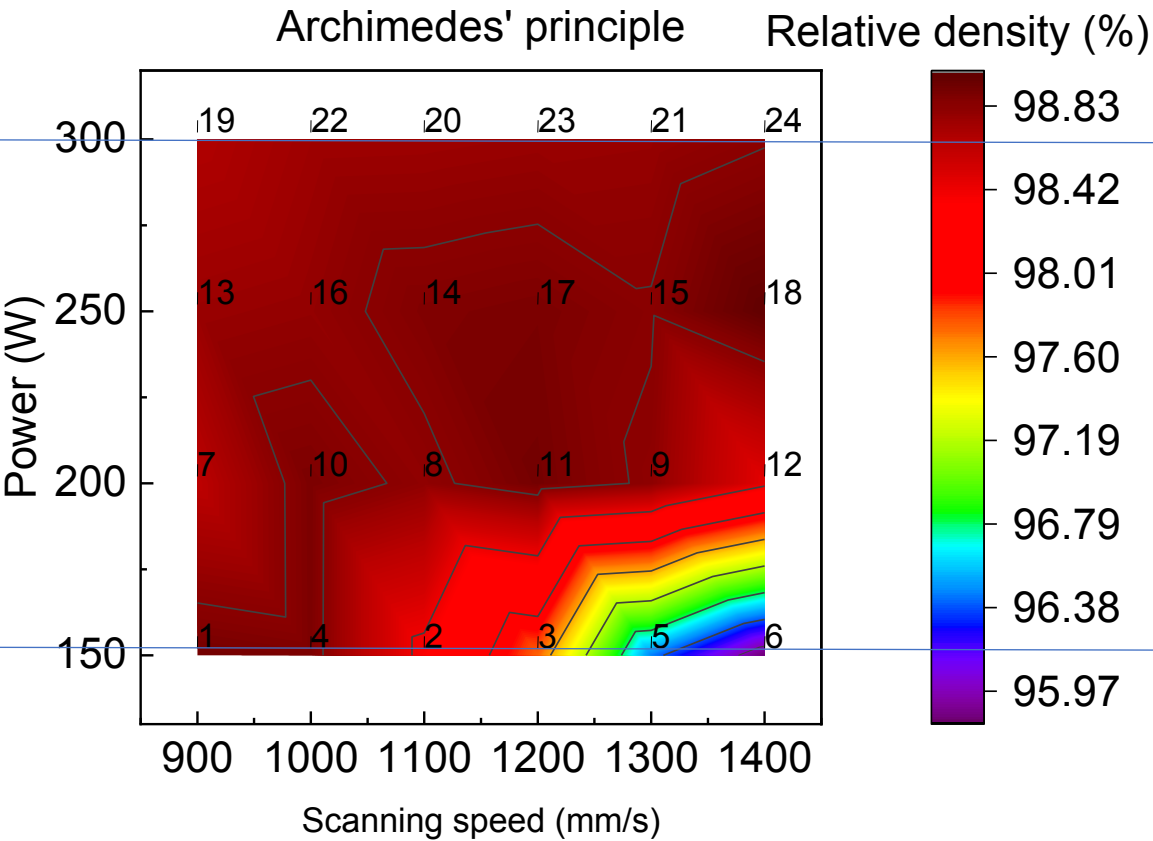
“Fully dense” zone

A steady relation between power and scanning speed is achieved and it is possible to obtain pieces with densities higher than 98 %.

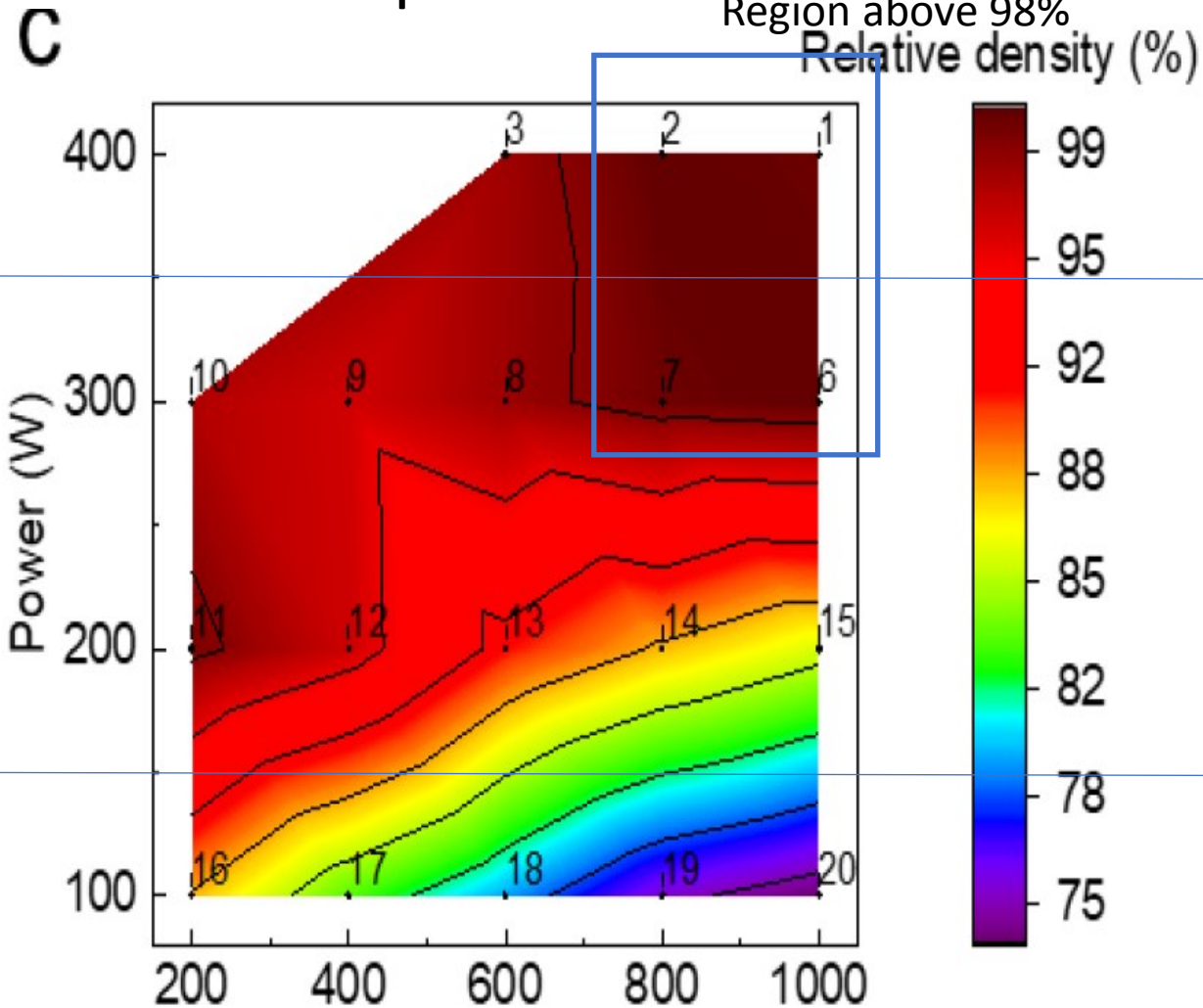
Samples from HDH powder

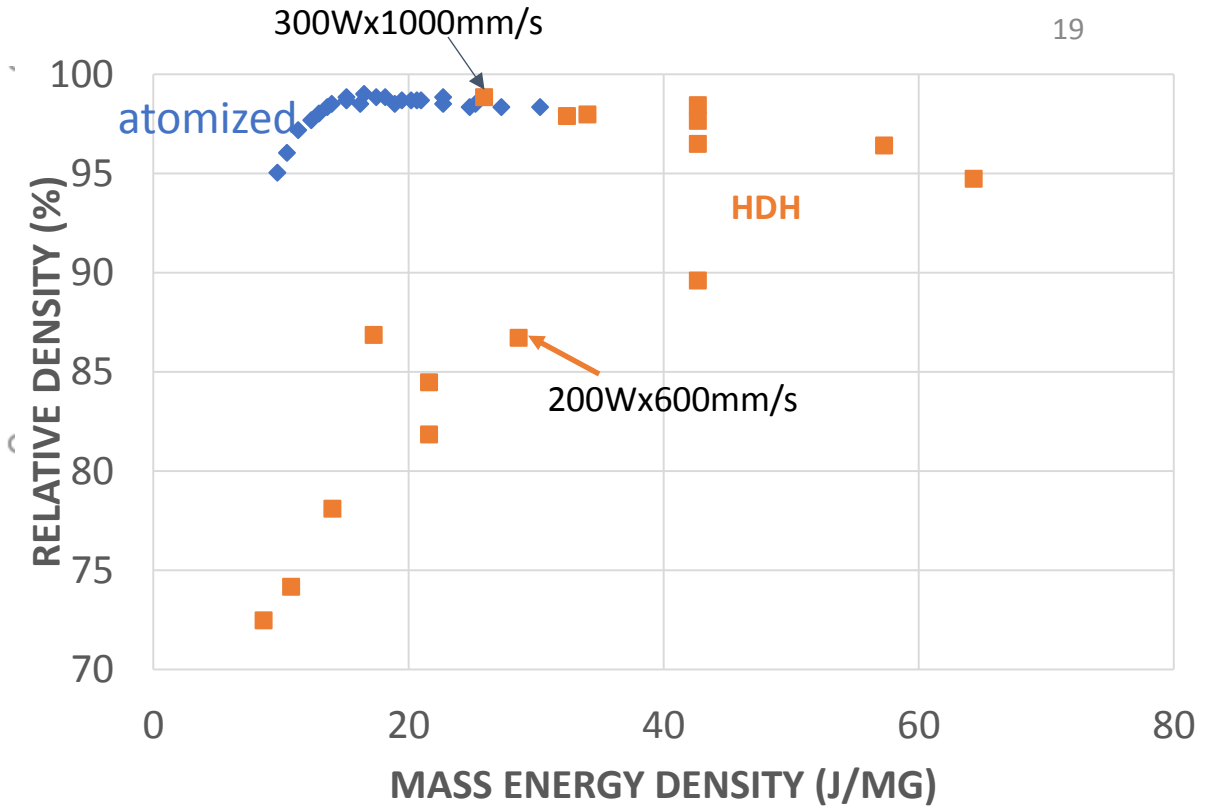
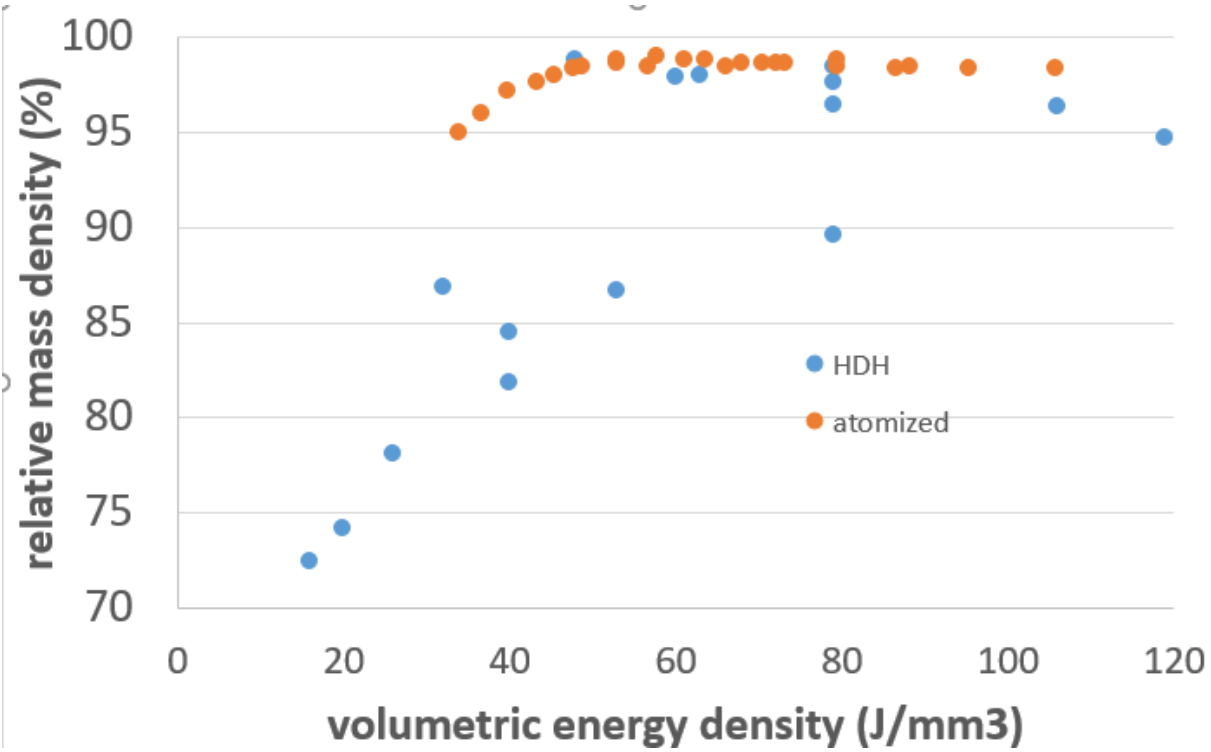


Atomized powder



HDH powder





Two different approaches to the energy density

Relative density was sensitive to the combination of power and speed. Low power was insufficient to promote densification, even when speed was decreased.

More detailed comparison

SLM parameters of fabrication:

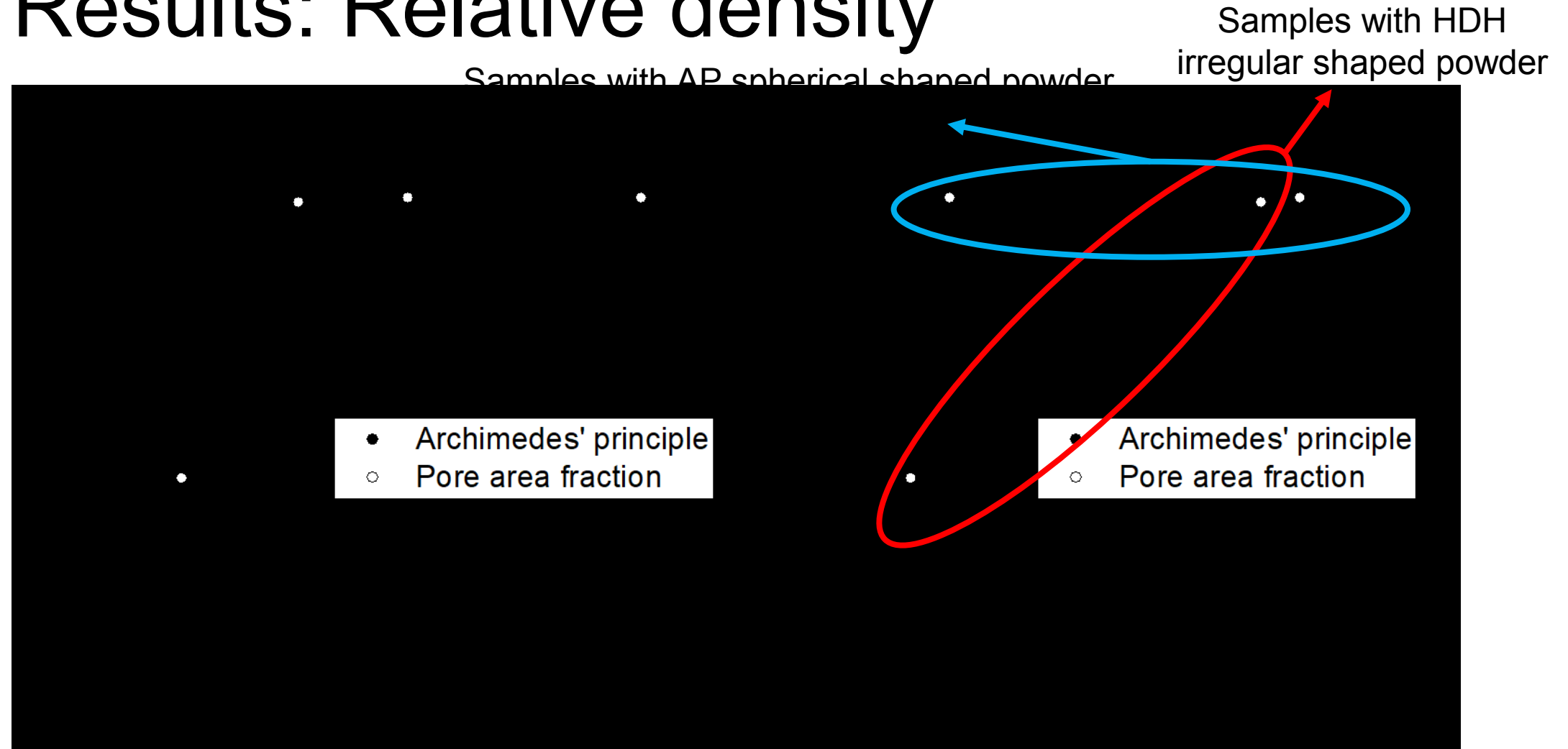
Same laser speed in four samples, comparing two powders at two laser power.

Sample	1AP	2AP	3HDH	4HDH
P_L (W)	200	300	200	300
V_S (mm/s)	1000	1000	1000	1000
H_S (μm)	105	105	105	105
D_S (μm)	30	30	60	60
E_V (J/mm ³)	63	95	32	48
E_M (J/mg)	18	27	17	26

For samples manufactured with AP powder

For samples manufactured with HDH powder.

Results: Relative density

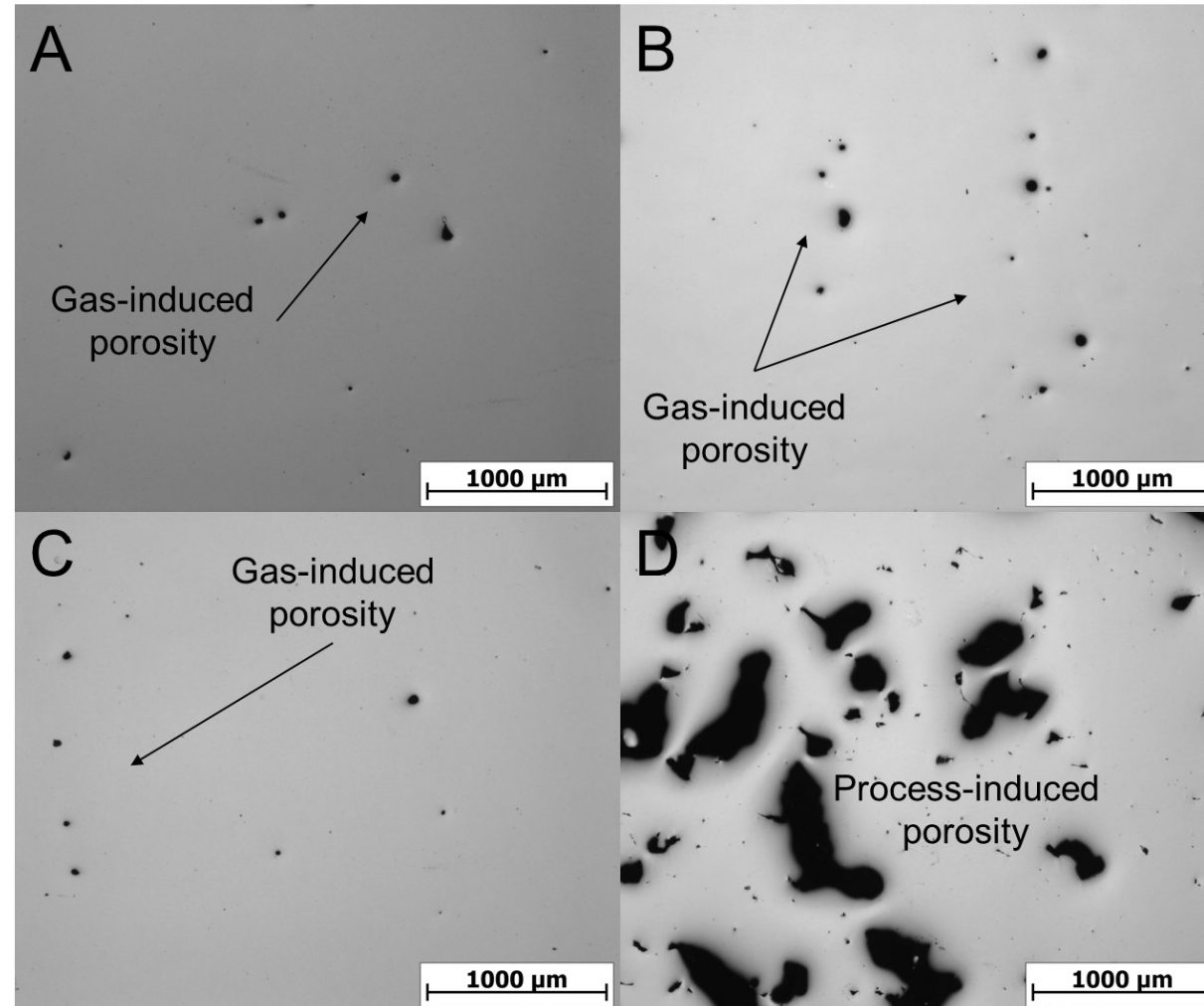


Values of relative density as a function of A) energy density and B) energy mass density.

Results: Microstrutural analysis

300 W
Samples
fabricated with
AP spherical
shaped powder

200 W

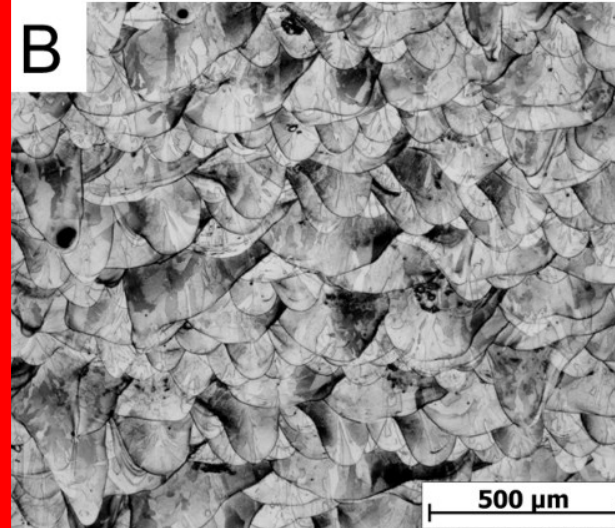
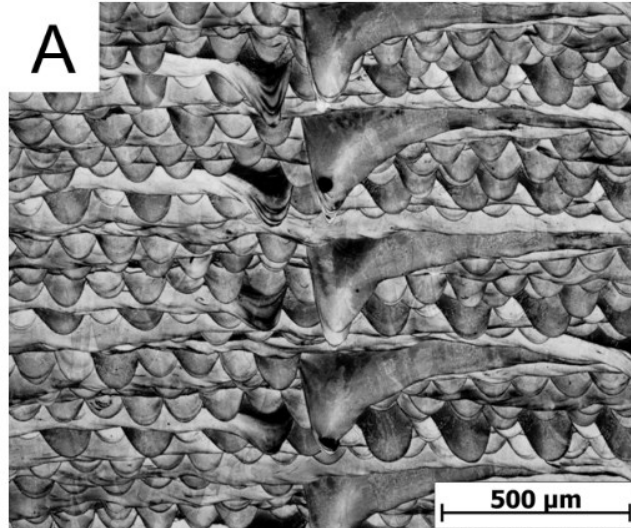


samples
fabricated with
HDH irregular
shaped powder

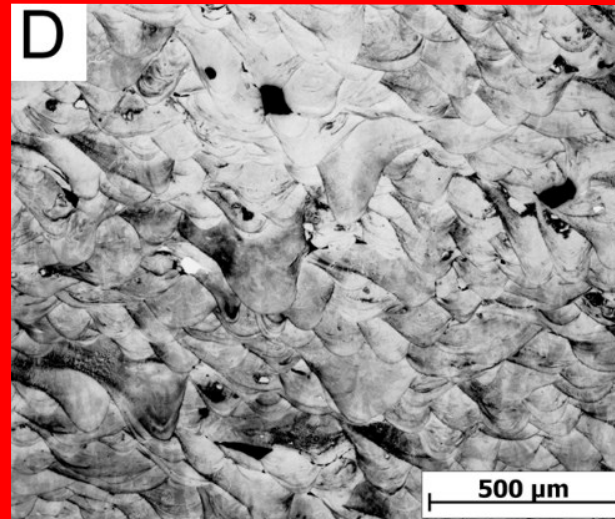
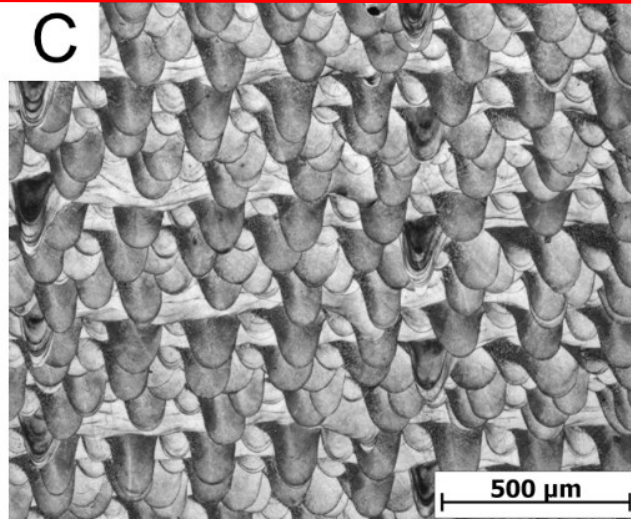
AP spherical
shaped powder

HDH irregular
shaped powder

300 W



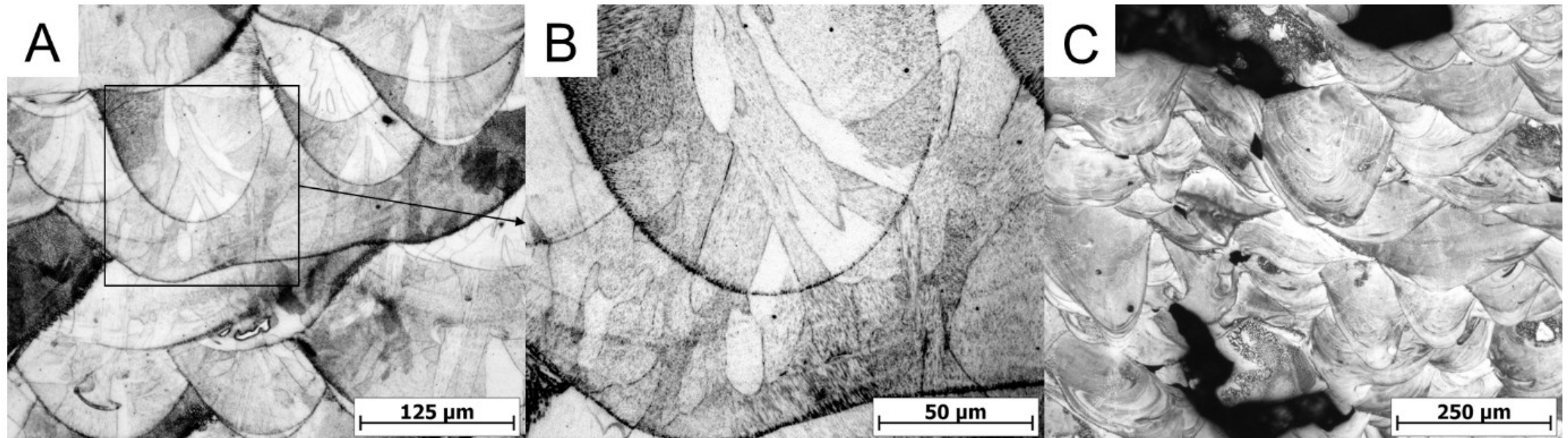
200 W



Results: Samples from HDH powder

300 W

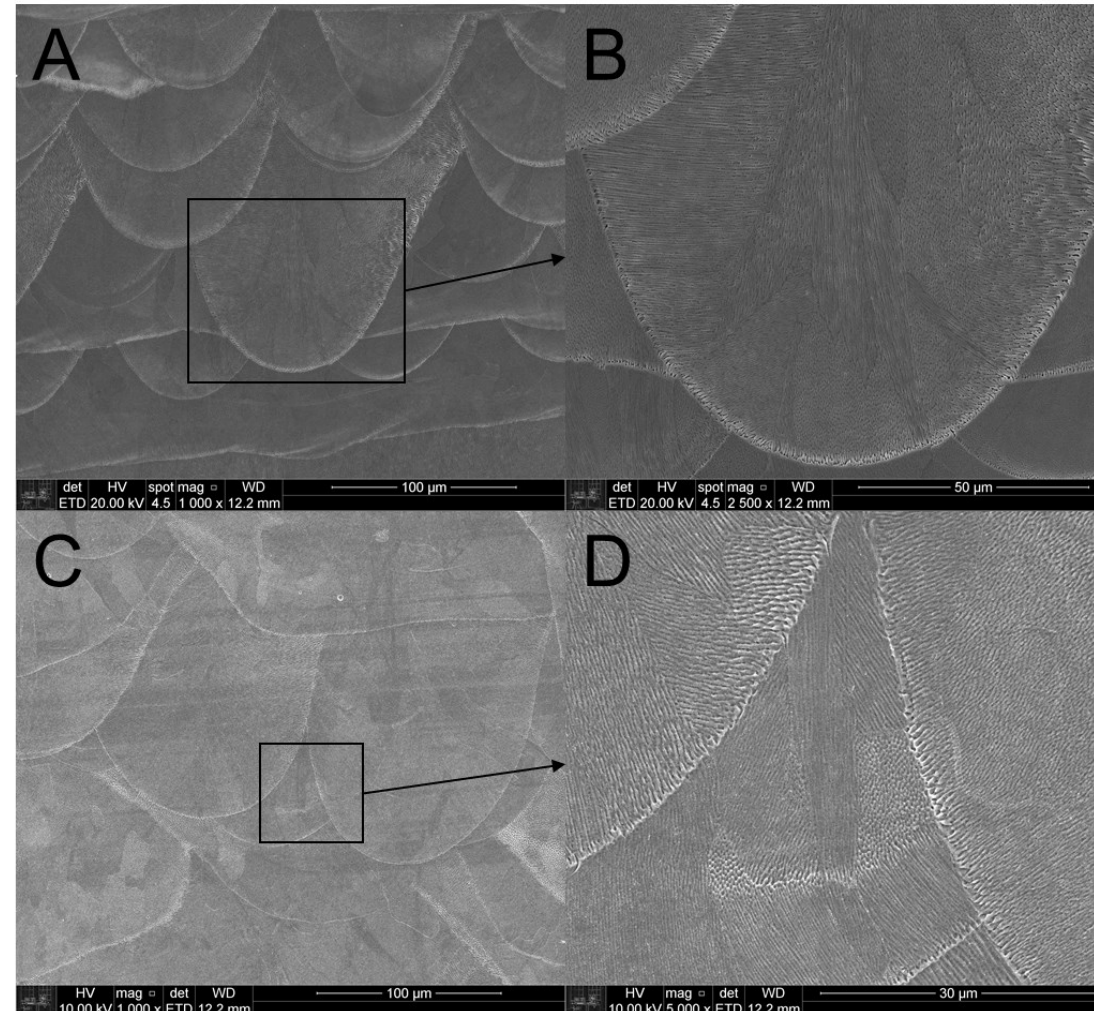
200 W



Results: Microstrutural analysis

Sample 2AP
fabricated with
AP spherical
shaped powder

Sample 4HDH
fabricated with
HDH irregular
shaped powder

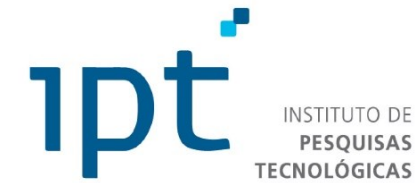


300 W

Conclusions

1. It has been possible to obtain density above 98%, with HDH powder, even though its apparent density is half of the atomized powder.
2. It was possible to obtain density above 98%, with atomized powder, even with laser power of 150W
3. Different processing windows were needed for HDH powder: Laser power above 200W was necessary for 98% densification.

Acknowledgment



Numero do processo 2016/50199-6 e 2018/02001-8



Thank you!

Fernando J.G. Landgraf

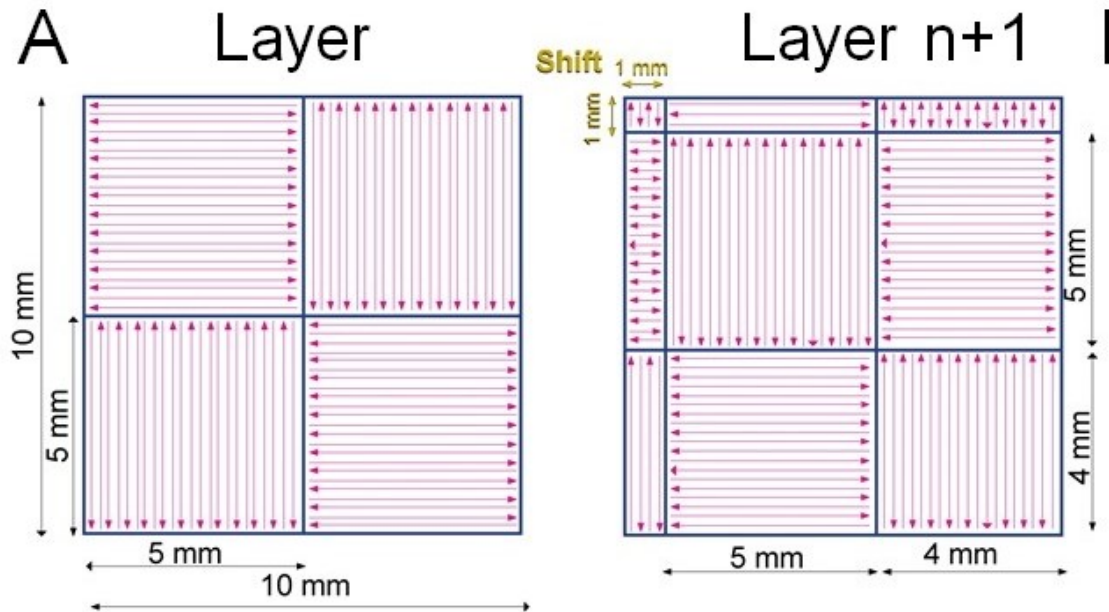
Polytechnic School of the University of São Paulo

Department of Metallurgical and Materials

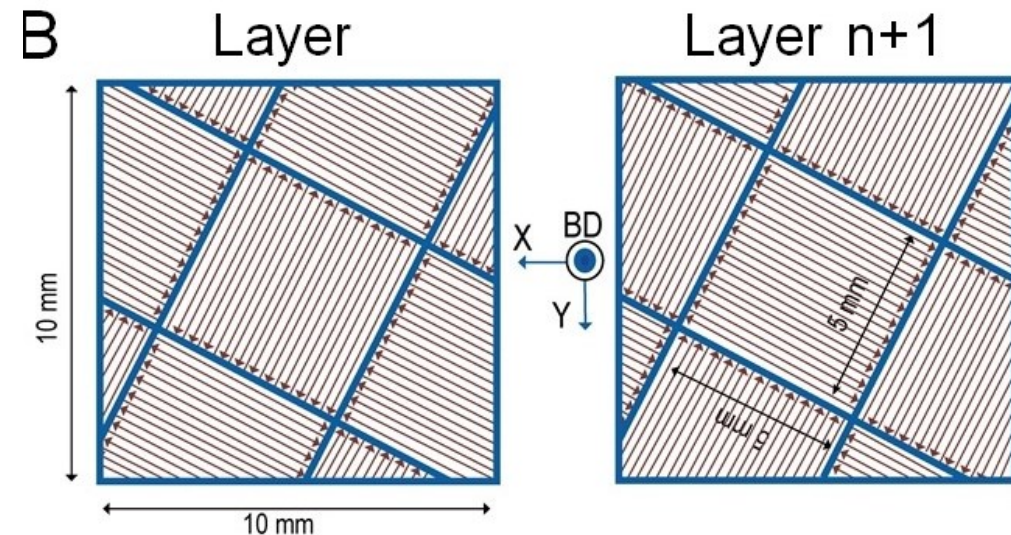
Engineering São Paulo, SP, Brazil

f.landgraf@usp.br

Some SLM parameters were changed



For samples manufactured with AP powder



For samples manufactured with HDH powder.

Content of oxygen inside the chamber

AP Spherical powder: Less than 0.1 %

HDH Irregular powder: between 0.4-0.6 %