

#### **COMUNICAÇÃO TÉCNICA**

#### N° 179727

Influence of microstrucute on the formation of corrosion products of steels susceptible to SCC-CO2

Guilherme Henrique Alves Andrade Carlos Alberto da Silva Zehbour Panossian

> Palestra apresentada no CONGRESSO INTERNACIONAL DE CORROSÃO, INTERCORR, 10., 2025, São Paulo. 24 slides

"Comunicação Técnica" compreende trabalhos elaborados por técnicos do IPT, apresentados em eventos, publicados em revistas especializadas ou quando seu conteúdo apresentar relevância pública. **PROIBIDO REPRODUÇÃO** 

Instituto de Pesquisas Tecnológicas do Estado de São Paulo S/A - IPT Av. Prof. Almeida Prado, 532 | Cidade Universitária ou Caixa Postal 0141 | CEP 01064-970 São Paulo | SP | Brasil | CEP 05508-901 Tel 11 3767 4374/4000 | Fax 11 3767-4099

www.ipt.br





Influence of Microstructure on the Formation of Corrosion Products of Steels Susceptible to SCC-CO<sub>2</sub>

**Guilherme Henrique Alves Andrade – FIPT/IPT** 





#### Authors:

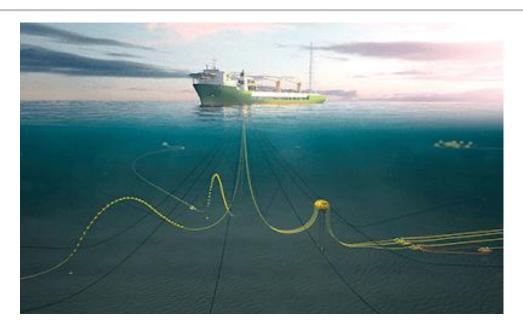
- Guilherme Henrique Alves Andrade FIPT/IPT eng.mec.guilhermeandrade@gmail.com
- Carlos Alberto da Silva FIPT/IPT
  callbertt@gmail.com
- Zehbour Panossian IPT zep@ipt.br

InterCorr2025\_233

Slide 2/23









- The pressure and tensile armours are made of carbon steel.
- Mechanical properties and corrosion resistance are key factors in their performance under aggressive environments.

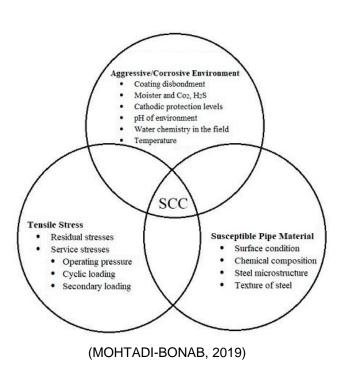


Slide 3/23





Among the most relevant degradation mechanisms in these systems is CO<sub>2</sub>-induced stress corrosion cracking (SCC-CO<sub>2</sub>).



Stress corrosion cracking is a **time-dependent process** in which a metal or alloy undergoes premature cracking due to the synergistic action of a corrosive environment and static tensile stresses—whether applied or residual. Neither the corrosive environment nor the stress alone is sufficient to cause SCC.

A technically significant incident was reported by the Brazilian National Agency of Petroleum, Natural Gas and Biofuels (ANP) in 2017: a flexible gas injection riser failed due to CO<sub>2</sub>-SCC just two years after installation, despite a projected service life of **20 years**. The failure, which occurred in the tensile armor wires, was caused by the ingress of water and CO<sub>2</sub> into the annular space of the riser.

Slide 4/23





In environments containing CO<sub>2</sub>, the formation of protective iron carbonate (FeCO<sub>3</sub>) layers on the surface of carbon steels is common. These layers act as barriers that reduce the contact between the metallic substrate and the aggressive environment, thereby contributing to the mitigation of generalized corrosion.

$CO_{2(aq.)} + H_2O \rightleftharpoons H_2CO_3$	Reaction 1
$H_2CO_3 + e^- \rightleftharpoons HCO_3^- + H_{(ads)}$	Reaction 2
$HCO_3^- + e^- \rightarrow CO_3^{2-} + H_{(ads)}$	Reaction 3
$Fe \rightarrow Fe^{2+} + 2e^{-}$	Reaction 4
$Fe^{2+} + CO_3^{2-} \rightarrow FeCO_3$	Reaction 5

The effectiveness of this protection depends on factors such as the morphology, thickness, adhesion, and integrity of the layer. Adhesion, in particular, plays a key role in the stability of the layer under mechanical stresses: poorly adhered films tend to delaminate, exposing the substrate to corrosive attack.

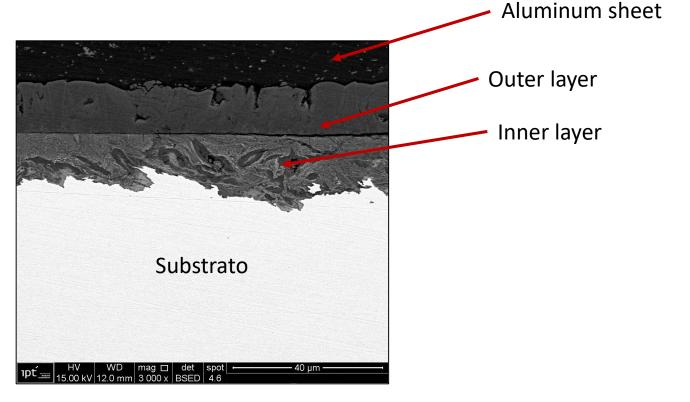
Slide 5/23





Siderite formation on carbon steels occurs in two main morphologies:

- an outer crystalized layer and
- an inner compact layer

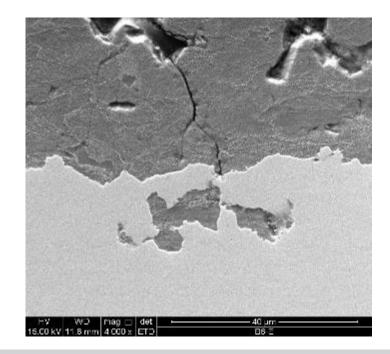


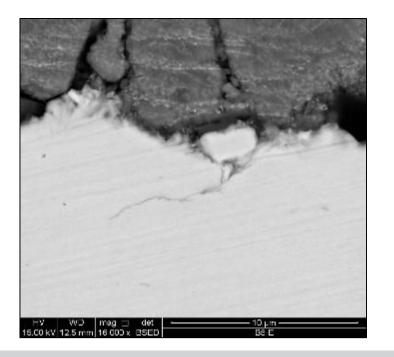






In the CO<sub>2</sub>-SCC failure mechanism, localized rupture of the FeCO<sub>3</sub> film allows for localized acidic attack, promoting crack nucleation. Additionally, the deformation imposed to elongate the specimen may cause the corrosion product layer to rupture at random locations, exposing the underlying metal to the electrolyte, which is the necessary condition for stress corrosion cracking to occur (PIMENTEL et al., 2025).





Slide 7/23







### **OBJECTIVE**



The objective of the present study is to evaluate the influence of the microstructure of different carbon steels on the formation of iron carbonate (FeCO<sub>3</sub>) layers in environments containing carbon dioxide ( $CO_2$ ).

Slide 8/23





Five carbon steels commonly used as tensile armor wires in flexible pipes were selected. All exhibit a ferritic-pearlitic microstructure, although with some key differences:

**Steel A**: low-carbon steel (0.3 %) with a predominantly ferritic microstructure and the presence of spheroidized cementite.

**Steel B**: steel with a carbon content between 0.6 % and 0.7 %, featuring spheroidized cementite.

**Steel C**: steel with a carbon content between 0.6 % and 0.7 %, predominantly composed of lamellar pearlite.

**Steel D**: steel with a carbon content between 0.73 % and 0.77 %, containing spheroidized cementite.

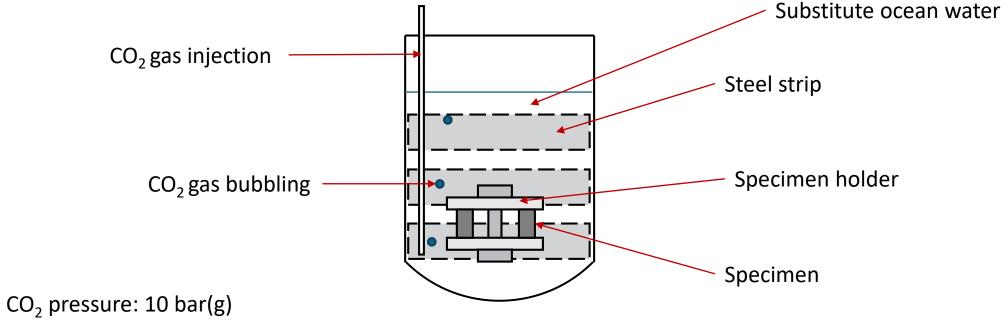
**Steel E**: steel with a carbon content between 0.73 % and 0.77 %, primarily composed of lamellar pearlite.

Slide 9/23





#### **Autoclave test setup**



Temperature: 40 °C

Relation volume/area (V/A): 1 mL/cm<sup>2</sup>

Duration: 47 days

Slide 10/23







The specimens were ground, finishing with **#600 grit** sandpaper, ensuring that the grinding marks were aligned with the drawing direction of the specimens — matching the direction of the surface marks produced during the wire drawing process.

For characterization, the following procedures were performed:

- Tensile tests: three tests per material;
- Chemical composition determined by optical emission spectrometry (OES);
- Microstructure evaluated by scanning electron microscopy (SEM).



Slide 11/23







Due to the cross section geometry of the wires, the areas marked in green, as schematically shown below, were selected for evaluation. SEM was used to measure the thickness of all iron carbonate layers (inner, outer, and total).



For each evaluated area, three regions were selected (one central and two at the edges) for measurements, with five measurements performed in each region. Thus, for each specimen, a total of 30 measurements were conducted per layer.

Slide 12/23



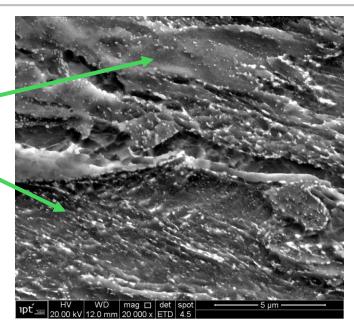




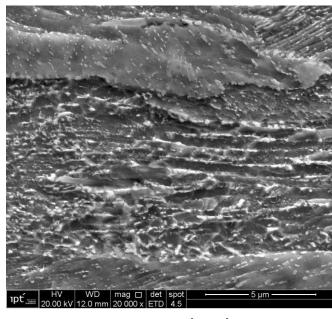
#### Steel A

**Ferrite** 

Spheroidized cementite



**Cross-section** 



Longitudinal

Mat	erial	Modulus of Elasticity [GPa]	Yield strength, 0.2 % [MPa]	Ultimate tensile strength [MPa]	Total elongation at fracture [%]
Charl A	Mean	169	833	949	12.4
Steel A	Standard Dev.	10	26	11	0.9

Matarial						Co	ontent (%wt)						
Material	С	Mn	Si	P	S	Cr	Ni	Мо	Al	Cu	Ti	V	Nb
Steel A	0.330	0.731	0.226	0.011	<0.001	ND	0.001	0.001	0.040	0.004	0.001	0.001	ND



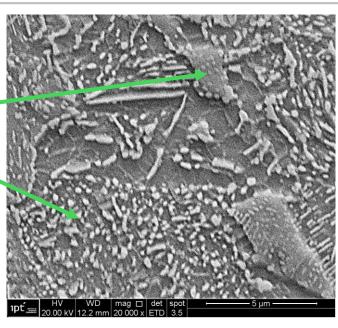




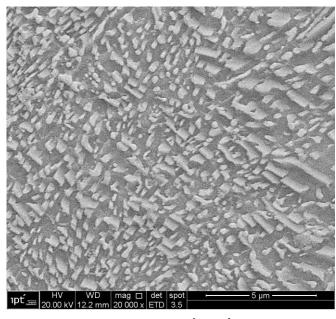
Steel B

Ferrite

Spheroidized cementite



**Cross-section** 



Longitudinal

Mat	erial	Modulus of Elasticity [GPa]	Yield strength, 0.2 % [MPa]	Ultimate tensile strength [MPa]	Total elongation at fracture [%]
Chool D	Mean	169	753.2	797.1	15.5
Steel B	Standard Dev.	22	4.4	3.2	1.0

Matarial		Content (%wt)												
Material	С	Mn	Si	Р	S	Cr	Ni	Мо	Al	Cu	Ti	V	Nb	
Steel B	0.602	0.650	0.280	0.006	<0.001	ND	0.001	0.001	0.041	0.001	0.001	0.002	ND	

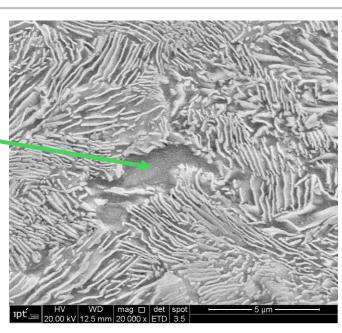




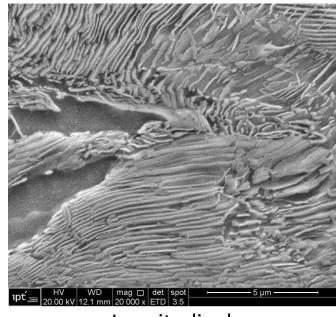
### Steel C

**Ferrite** 

Lamellar pearlite



**Cross-section** 



Longitudinal

Mat	erial	Modulus of Elasticity [GPa]	Yield strength, 0.2 % [MPa]	Ultimate tensile strength [MPa]	Total elongation at fracture [%]
Shool C	Mean	180.6	1337	1506	4.19
Steel C	Standard Dev.	1.5	68	69	0.25

Matarial	Content (%wt)												
Material	С	Mn	Si	Р	S	Cr	Ni	Мо	Al	Cu	Ti	V	Nb
Steel C	0.682	0.661	0.229	0.004	<0.001	ND	0.001	0.001	0.030	0.007	0.001	0.003	ND



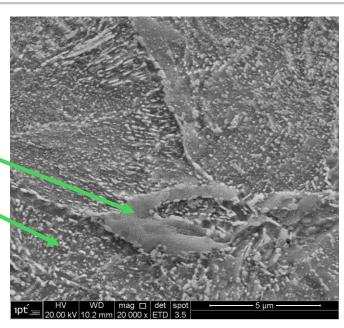




### Steel D

Ferrite

Spheroidized cementite



**Cross-section** 



Longitudinal

Mat	erial	Modulus of Elasticity [GPa]	Yield strength, 0.2 % [MPa]	Ultimate tensile strength [MPa]	Total elongation at fracture [%]
Charl D	Mean	222	1160.3	1273.4	9.4
Steel D	Standard Dev.	16	4.9	3.8	1.1

Matarial	Content (%wt)												
Material	С	Mn	Si	Р	S	Cr	Ni	Мо	Al	Cu	Ti	V	Nb
Steel D	0.732	0.811	0.241	0.008	<0.001	ND	0.001	0.001	0.029	0.008	0.001	0.002	ND

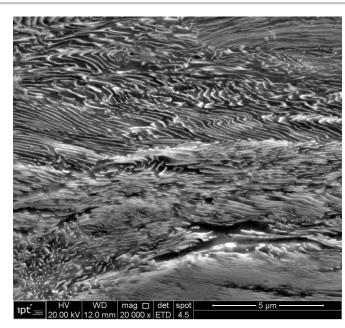




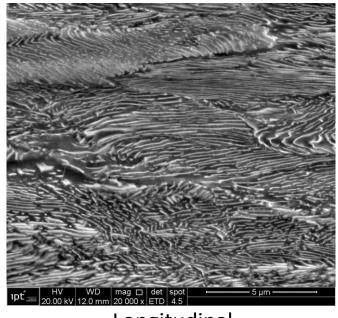


#### Steel E

Predominantly lamellar pearlite



**Cross-section** 



Longitudinal

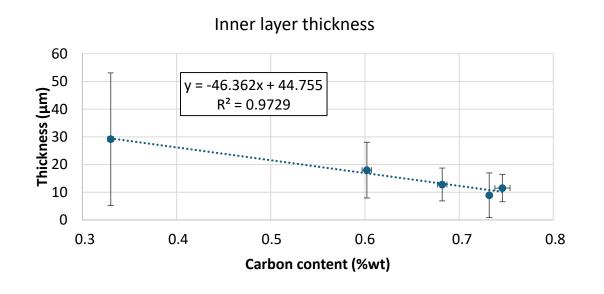
Mat	erial	Modulus of Elasticity [GPa]	Yield strength, 0.2 % [MPa]	Ultimate tensile strength [MPa]	Total elongation at fracture [%]
Ctool F	Mean	166	1300	1505	5.1
Steel E	Standard Dev.	70	30	40	1.3

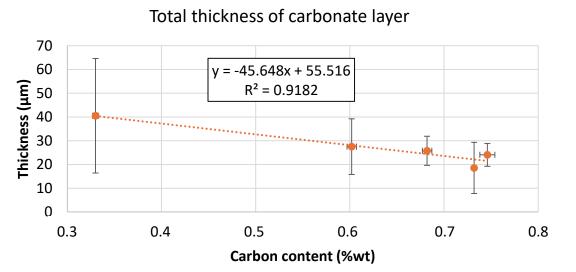
	Matarial	Content (%wt)								ówt)					
Mate	Material	С	Mn	Si	Р	S	Cr	Ni	Mo	Al	Cu	Ti	V	Nb	
	Steel E	0.746	0.775	0.278	0.013	0.015	0.028	0.001	<0.002	0.001	0.024	0.002	0.002	0.002	





#### Relationship between carbon content and the inner layer thickness



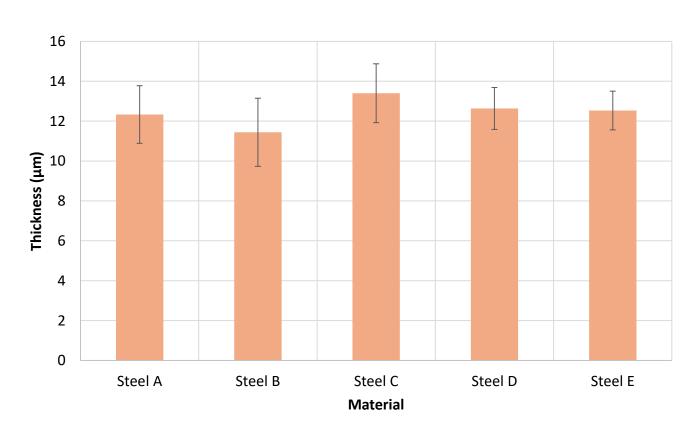


- There is a tendency for the thickness of the inner carbonate layer to decrease as the carbon content increases.
- The standard deviation values also tend to decrease with increasing carbon content.





#### **Outer layer thickness**



According to the analysis of variance (ANOVA), the values are statistically different (p-value of  $2.65 \times 10^{-5}$ ), with no clear trend in the measurement behavior.

However, when compared with the thickness values of the inner layer, this difference becomes negligible.

In the spheroidized steels, regions were identified where the outer layer was not present.

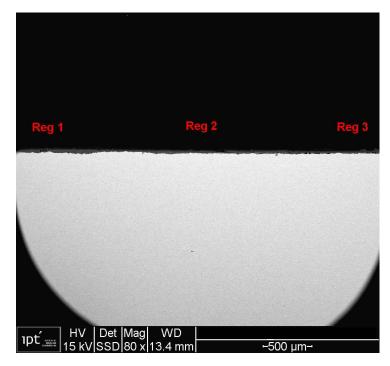
Slide 19/23

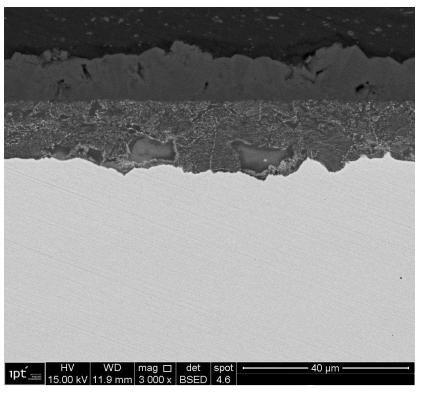


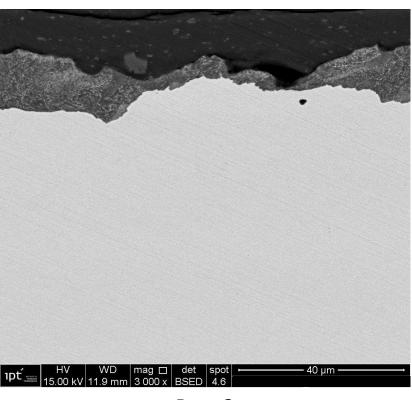




### Steel B







Reg. 2

Reg. 3

During the test, the outer layer detached from the specimen surface

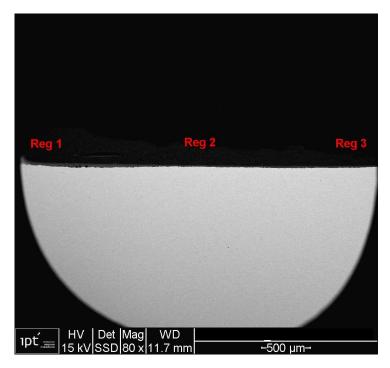
Slide 20/23

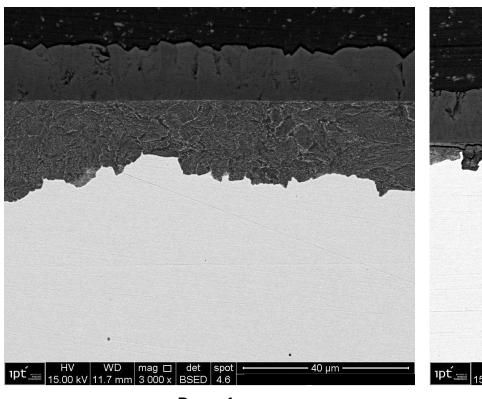


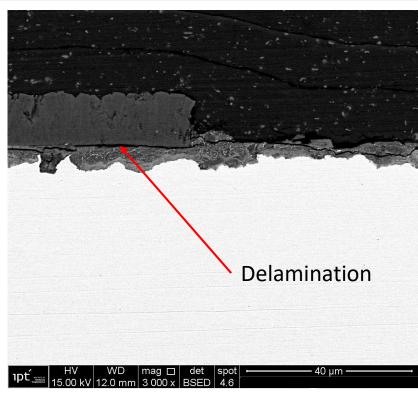




#### Steel D







Reg. 1 Reg. 2

The spheroidized microstructure provided poorer mechanical support for the outer layer, leading to adhesive failure.

Slide 21/23







### CONCLUSIONS



- The effect of carbon content on the thickness of the outer iron carbonate layer showed statistically significant differences according to ANOVA; however, when compared to the total thickness of the complete carbonate layer, its effect is negligible.
- An increase in the carbon content of the evaluated hypoeutectoid steels led to a reduction in the thickness of the inner iron carbonate layer.
- The outer carbonate layer of steels with a higher proportion of spheroidized pearlite provided poorer mechanical support, as evidenced by the presence of cracks and delamination.



### **FUTURE WORK**



#### Adhesion evaluation

- Instrumented nanoindentation tests
- Continuous scratch tests to assess adhesion between layers

#### **Carbon content effect**

Repeat the same tests on lower carbon steels (e.g., steels used in rigid pipes)

#### **Environmental parameters**

 Study the effect of CO₂ partial pressure and exposure time on the morphology and properties of iron carbonate layers





# THANK YOU!